Effects of proton irradiation on flux-pinning properties of underdoped $\text{Ba}(\text{Fe}_{0.96}\text{Co}_{0.04})_2\text{As}_2$ pnictide superconductor

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Abstract

We study the effect of proton irradiation on $\text{Ba}(\text{Fe}_{0.96}\text{Co}_{0.04})_2\text{As}_2$ superconducting single crystals from combined magnetisation and magnetoresistivity measurements. The study allows the extraction of the values of the apparent pinning energy U_0 of the samples prior to and after irradiation, as well as comparison of the values of U_0 obtained from the flux-flow reversible region with those from the flux-creep irreversible region. Irradiation reduces T_c modestly, but significantly reduces U_0 in both regimes: the critical current density J_c is modified, most strikingly by the disappearance of the second magnetisation peak after irradiation. Analysis of the functional form of the pinning force and of the temperature dependence of J_c for zero field, indicates that proton irradiation in this case has not changed the pinning regime, but has introduced a high density of shallow point-like defects. By consideration of a model that takes into account the effect of disorder on the irreversibility line, the data suggests that irradiation produced a considerable reduction in the average effective disorder overall, consistent with the changes observed in U_0 and J_c .

Keywords: pnictides, proton irradiation, superconducting critical current, flux-pinning

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1. Introduction

Since the discovery of the iron pnictides superconductors [1], flux-pinning has been widely studied in most of the iron-superconducting systems, in order to evaluate the potential of this material for application [2, 3, 4, 5]. As previously studied in copper based high- T_c superconductors [6, 7, 8, 9], the study of flux-pinning after proton and particle irradiation in pnictides has gained increasing interest [10, 11, 12, 13, 14, 15, 16, 17, 18, 19]. Irradiation allows the effect of disorder on the vortex state to be studied by changing the flux-pinning, leading in some cases to the disappearance of the second magnetisation peak, SMP, (often described as the fishtail peak) [10]. Depending on the irradiation dose, it has been shown that the effect of disorder can reduce the value of T_c and potentially also change the Fermi level in the bands participating in the pairing mechanism.[20]

In this work, we study the effect of proton irradiation on underdoped $\text{Ba}(\text{Fe}_{0.96}\text{Co}_{0.04})_2\text{As}_2$ single crystals [21, 22]. For $\text{Ba}(\text{Fe}_{1-x}\text{Co}_x)_2\text{As}_2$ system, the underdoped content of Co occur for $x < 0.06$ [22]. The investigation presents a comparative study performed on two crystals from the same batch, with the same superconducting transition temperatures, $T_c = 13$ K, after which one of the crystals was irradiated, producing a drop in the value of the transition temperature, to $T_c \sim 12$ K and the disappearance of the SMP. The study was conducted by obtaining isothermal resistivity curves as a function of magnetic field and isothermal magnetisation curves as a function of magnetic field and time (isofield magnetic relaxation curves) on both crystals. Results of the analysis allowed the comparison of the effect of irradiation on the activation energy, U_0 , as well as the comparison for each sample, of the values of U_0 obtained from resistivity taken in the reversible regime, with the U_0 values obtained from flux-creep, in the irreversible regime. We also study the effect of irradiation on the normalized volume pinning force, on the temperature dependent behavior of the normalized remanent critical current, and on the irreversible lines which were fitted with

an expression[23] that allows the estimation of the disorder within each sample.

2. Experimental

The high-quality $\text{Ba}(\text{Fe}_{0.96}\text{Co}_{0.04})_2\text{As}_2$ single crystals used in this work with approximate dimensions: crystal 1 (virgin) $2.7 \times 2.2 \times 0.19 \text{ mm}^3$ and mass m
45 $= 5.72 \text{ mg}$, crystal 2 (irradiated) $0.9 \times 0.8 \times 0.025 \text{ mm}^3$ and mass $m = 0.85 \text{ mg}$ and crystal 3 (irradiated) $0.9 \times 0.75 \times 0.02 \text{ mm}^3$ and mass 0.96 mg , were grown by the CoAs flux method [21]. Crystal 3 was only studied by electrical transport and was damaged after a unique isothermal $M(H)$ at $T = 6 \text{ K}$ was obtained. The crystals prior to irradiation exhibit a $T_c = 13 \text{ K}$ with $\delta T_c <$
50 1 K . Proton irradiation has been shown to introduce defects, which are recognized to be predominantly of a point defect character, without significantly altering the intrinsic electronic structure of the material [24, 25]. Proton irradiation induces charge carrier scattering, increasing the residual electrical resistivity $\delta\rho(0)$, and lowering both T_N , the Neel temperature transition, and T_c
55 [24, 26]. In the present study proton irradiation was performed such that a proton dose of $0.5 \times 10^{16} \text{ cm}^{-2}$ (crystal 3) and $1.5 \times 10^{16} \text{ cm}^{-2}$ (crystal 2) were achieved for the $\text{Ba}(\text{Fe}_{0.96}\text{Co}_{0.04})_2\text{As}_2$ crystals. Crystal sizes were sufficiently thin ($< 50 \text{ }\mu\text{m}$) to achieve homogeneous irradiation damage using 3-MeV H^+ irradiation according to calculations using the SRIM code. The observed ratio
60 of $(\delta T_c / \delta\rho(0))$ with irradiation, produced a value of $-0.02 \text{ K}(\mu\Omega\text{cm})^{-1}$ which is significantly lower than the $-0.08 \text{ K}(\mu\Omega\text{cm})^{-1}$ calculated by Nakajima et al [24] on $\text{Ba}(\text{Fe}_{0.955}\text{Co}_{0.045})_2\text{As}_2$. In the present work, the irradiation was conducted at room temperature while in comparable studies low temperature proton irradiation (50 K) was utilized. Taen et al [27] have suggested that the random
65 point defects generated by the irradiation may be annealed out of the structure at room temperature. This possibility is unlikely to have occurred in the present study as the samples were properly thermally anchored by mounting on a large copper block during irradiation avoiding any possible overheating. This is corroborated by the increase of residual resistivity and decrease of T_c and T_N

70 (as shown below when discussing Fig. 1) indicating that additional defects have been consistently created by the irradiation.

All transport measurements were taken using the van der Pauw technique [28] with the magnetic field applied along the c axis and current within the ab plane. Magnetisation measurements as a function of temperature, $M(T)$, and
75 applied field, $M(H)$, were made with a Quantum Design VSM-9T system. The results were obtained after cooling the crystals from above T_c to the desired temperature in zero applied magnetic field (ZFC), after which the magnetic field is applied along the c -axis. Magnetic hysteresis loops were obtained for various temperatures below T_c in each crystal, and magnetic relaxation curves
80 were obtained for selected magnetic fields and temperatures over span times extending up to 3 hours.

3. Results and discussion

Figure 1 shows a plot of resistivity as a function of temperature for an unirradiated crystal and two other irradiated crystals after exposed to different
85 doses. As already mentioned, both T_N and T_c are lowered after irradiation, which is shown in an inner inset of Fig. 1. The values of T_c in this inset were obtained by the 80 % value of the normal resistivity criterion. A second inset in Fig. 1 shows the effect of irradiation on isothermal $M(H)$ curves obtained at $T = 6$ K. The data on crystal 3 is included only to illustrate the effect of irradiation
90 on T_c on T_N and on the $M(H)$ curves. Figure 1 and the insets show that the low dose of proton irradiation reduced both the value of T_c and of the transition width $\delta T_c/T_c$, as observed in optimally doped $\text{Ba}(\text{Fe}_{0.93}\text{Co}_{0.07})_2\text{As}_2$ [19] ($T_c = 24$ K), but produced an overall decrease of the magnetization values when compared with the unirradiated sample. The highest dose of irradiation reduced
95 even further the value of T_c but increased the transition width. This apparent unusual behavior was also observed in underdoped samples of $(\text{Ba-K})\text{Fe}_2\text{As}_2$ after high doses of proton irradiation [16]. The highest dose of irradiation also produced an additional, consistent, decrease of the values of magnetization, as shown in the right inner inset.

Figure 1: Resistivity curves for three different crystals after different doses of proton irradiation. Insets: T_C and T_N as a function of irradiation doses; isothermal $M(H)$ curves at $T = 6 \text{ K}$ as a function of irradiation doses.

Figure 2 shows the effect of irradiation on the estimated critical current density at $T = 3 \text{ K}$. The critical current density is given by [29] $J_c = 20\Delta M / (a(1 - a/3b))$, where ΔM is the full width of each hysteresis curve shown in the upper inset of Fig. 2 and $b > a$ are the dimensions of the samples. Arrows in the inset figure marks the region where the second magnetisation peak, SMP, develops in the virgin sample and shows the irreversible field, H_{irr} , for the irradiated sample. As shown in Fig. 2, irradiation produces a large reduction of J_c and the disappearance of the SMP.

To further study the effect of irradiation on flux pinning we perform magnetic

Figure 2: Isothermal $J_c(H)$ curves as a function of magnetic field at $T = 3 \text{ K}$. inset: isothermal magnetisation versus field hysteresis curves as obtained for crystal 1 and 2 at $T = 3 \text{ K}$.

relaxation measurements (M vs. time) on both crystals at fixed temperatures
 110 and fixed applied magnetic fields. Figure 3a shows selected isofield $\ln M$ vs
 $\ln t$ curves obtained at $T = 3 \text{ K}$ for the virgin sample. Logarithmic behavior
 with time was observed in all relaxation curves, on both crystals, allowing the
 apparent activation energy U_0 defined by $d \ln M / d \ln t = k_B T / U_0$, to be obtained.
 Figure 3b shows isofield plots of the logarithm of the resistivity as a function
 115 of $1/T$ as extracted from isothermal magnetoresistivity curves obtained for the
 irradiated sample. Similar plots were obtained for the virgin sample following
 the same procedure. As shown in the curves of Fig. 3b, an approximately

Figure 3: a) Example magnetic relaxations curves as obtained for the virgin sample; b) Isofield magnetoresistivity curves as obtained for the irradiated sample.

linear behavior of $\ln R$ occurs in the lower region of the curves, which appears to be associated with the well known thermally assisted flux flow resistivity [30], where $\ln R = \ln R_0 - U_0/k_B T$, allowing the flux-flow activation pinning energy U_0 to be estimated for each curve.

Figure 4 shows a plot of the activation pinning energy U_0 as estimated from flux-creep data in the irreversible regime and from magnetoresistivity data in the reversible regime for both samples. The plots shown in Fig. 4 allow one to compare and to observe that values of U_0 obtained from flux-creep (in the irreversible regime) are quite different and lower than values obtained in the flux-flow regime. As well as the difference between the activated pinning energy within the different regimes, it is also clear that irradiation significantly reduces U_0 . This result is rather interesting, motivating us to further explore the influence of irradiation on the pinning force and the pinning mechanism.

Figure 5 shows plots of several isothermic curves of the normalised volume pinning force, $F_p/F_{p_{max}}$, as a function of the reduced field $h = H/H_{irr}$ as obtained for both samples, where $F_p = J_c \times B$ and B is the induction magnetic field (it is assumed that $B = H$ which is a good approximation for fields far from the Meissner region). It should be noted that the many different curves in each figure appear to collapse to a single curve. Solid lines appearing on Fig.

Figure 4: U_0 as obtained from flux-flow and from flux-creep on both samples is plotted against the magnetic field. Dotted lines are guide-for-the eye only.

5a (virgin crystal) and Fig. 5b (irradiated crystal) are obtained by fitting the data to the expression $Fp \approx h^p (1-h)^q$ [31] where p and q are fitting parameters which provide insight into the type of pinning. Moreover $p/(p+q)$ represents the field, h_{max} , for which the maximum value of Fp/Fp_{max} occurs. Results of the fittings are: $p = 0.5$ and $q = 3.4$ for the virgin crystal (Fig. 5a) and $p = 0.36$ and $q = 2$ for the irradiated sample (Fig. 5b). While the values of p and q for the virgin crystal are difficult to explained in terms of possible values expected for different types of pinning [31], the values of p and q for the irradiated sample suggest that point pinning dominates in this case. It is interesting to note that

Figure 5: The normalised pinning force is plotted against the reduced field h for: a) virgin sample ; b) irradiated sample.

the value of h_{max} virtually did not change with irradiation.

It should be mentioned that the low value of h_{max} for both samples, suggests that pinning is of δ_L pin type (spatial variations in the mean free path of the carriers) for both samples. To check this finding, we plot in Fig. 6 values of the so called remanent state J_c for $H = 0$ (normalised by J_c at $T = 2$ K), $J_c/J_c(2K)$, as a function of the reduced temperature, T/T_c , which are obtained from the remanent magnetisation. It is interesting to observe that the normalised values of J_c for both samples almost superpose each other, suggesting that despite the differences observed in U_0 , T_c and J_c , due to irradiation, the same type of pinning is dominant in both samples. As performed in Ref. [32] we plot in Fig. 6 the values of $J_c/J_c(2K)$ calculated assuming that pinning is of δ_L pin type for which the temperature dependence is given by $(1-t^2)^{2.5}(1+t^2)^{-0.5}$ (red dotted line) and assuming that pinning is of δ_{T_c} pin type (random variation of T_c within the sample volume) which has a temperature dependence given by $(1-t^2)^{7/6}(1+t^2)^{5/6}$ (blue dashed line). A comparison between the curves suggests that δ_L pin is the predominant type of pinning for both samples, which agrees with the low value of h_{max} observed for both samples in Fig. 5.

Finally we present in Fig. 7 a plot of the temperature dependence of the irreversibility line, IL, H_{irr} vs T , for the studied samples, where values of H_{irr}

Figure 6: The normalised remanent critical current is plotted against temperature for both samples. Dotted and dashed lines represents the values expected for δ_L pin and δ_{T_c} pin.

are obtained by extracting from each hysteresis curve the field for which magnetisation become reversible. A simple inspection by eye shows the effect of irradiation on the IL. Since irradiation is usually associated with an increase disorder which in the present case appears in conflict with the large reduction of J_c as well U_0 after irradiation, it is interesting to fit the IL of Fig. 7 to an expression presented in ref. [23], which was developed considering a parameter that measures the disorder (n_p). Solid lines in the curves of Fig. 6 represents the best fit of each IL to the expression,

$$1 - t - b + 2[n_p(1 - t)^2 b / 4\pi][3/2 - (4\pi t \sqrt{2Gi} / (n_p(1 - t)^2))] = 0$$

where $t=T/T_c$ is the reduced temperature, $b=H/H_{c2}(0)$ is the reduced field, n_p is a parameter measuring the disorder and Gi is a parameter associated to the Ginzburg number measuring the strength of thermal fluctuations. The best fit
165 to the virgin curve is obtained assuming $T_c = 13$ K producing the values $H_{c2}(0) = 250$ kOe, $n_p = 0.005$ and $Gi = 5 \times 10^{-5}$. For the irradiated curve, the best fit is obtained assuming $T_c = 10$ K producing the values $H_{c2}(0) = 130$ kOe, $n_p = 0.0008$ and $Gi = 5 \times 10^{-5}$. Similar values of n_p , but a relatively lower value
170 of Gi were obtained for Ba(Fe-Ni)₂As₂ samples [33]. It should be noted that the fittings produced the same value of Gi for both curves, but interestingly a much smaller value of the disorder parameter n_p for the irradiated curve, which is consistent with the observed decrease in J_c as well as in U_0 after irradiation. This model suggests that irradiation creates a high density of defects with very
175 low activation pinning energy such that the overall effect is to reduce the average volume pinning and the effective disorder.

4. Conclusions

In conclusion, our study shows that proton irradiation produced many changes in intrinsic parameters of the superconductor Ba(Fe_{0.96}Co_{0.04})₂As₂. We find
180 that irradiation in this case produces an apparent healing of the crystal in that the critical current density, the extracted U_0 and the irreversibility line are all reduced after irradiation. We observe that values of U_0 obtained from flux creep are lower than those obtained from the thermal assisted flux flow region. Despite these changes, analysis based on the pinning force and on the temperature
185 dependence of J_c suggest that the same type of pinning, δ_L pin, remains after irradiation. One of the interesting observations is that the resulting defect character in the irradiated crystal causes the disappearance of the SMP, shedding light on the origin of this widely discussed feature.

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Figure 7: H_{irr} is plotted against the temperature for both samples. Dashed lines in each curve represents a fitting to the theory (see text for details).

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References

- [1] Y. Kamihara, T. Watanabe, M. Hirano and H. Hosono, *J. Am. Chem. Soc.* **130** 3296 (2008); H-H Wen, G. Mu, L. Fang, H. Huang and X. Zhu, *Europhys. Lett.* **82** 17009 (2008); Z-A Ren, J.Y. Yang, W. Lu, W. Lei, X-L Shen, Z-C Li, G-C Che, X-L Dong, L-L sun, F. Zhou and Z-X Zhao,

- Europhys. Lett.* **82** 57002 (2008); Y. Nakajima, T. Taen, and T. Tamegai, *J. Phys. Soc. Japan* **78** 023702 (2009)
- [2] D. C. Johnston, *Adv. Phys.* **59** 803 (2010); P.M. Aswathy, J.B. Anooja, P.M. Sarun and U. Syamaprasad, *Supercond. Sci. Technol.* **23** 073001 (2010)
- [3] C.J. Van Der Beek, M. Konczykowski, S. Kasahara, T. Terashima, R. Okazaki, T. Shibauchi and Y. Matsuda, *Phys.Rev. Lett.***105** 267002 (2010)
- [4] C. J. van der Beek, G. Rizza, M. Konczykowski, P. Fertey, I. Monnet, Thierry Klein, R. Okazaki, M. Ishikado, H. Kito, A. Iyo, H. Eisaki, S. Shamoto, M. E. Tillman, S. L. Budko, P. C. Canfield, T. Shibauchi, and Y. Matsuda., *Phys. Rev. B* **81** 174517 (2010)
- [5] A. Yamamoto, J. Jaroszynski, C. Tarantini, L. Balicas, J. Jiang, A. Gurevich, D. C. Larbalestier, R. Jin, A. S. Sefat, M. A. McGuire, B. C. Sales, D. K. Christen and D. Mandrus., *Appl. Phys. Lett.* **94** 062511 (2009)
- [6] D. Larbalestier, A. Gurevich, D.M. Feldman and A. Polyanski, *Nature* **414** 368 (2001)
- [7] W. Gerhauser, G. Ries, H.W. Neumuller, W. Schmidt, O. Eibl, Saemann-Ischenko and S. Klaumunzer, *Phys. Rev. Lett.* **68** 879 (1992)
- [8] M. Konczykowski, F. Rullier-Albenque, E.R. Yacoby, A. Shaulov, Y. Yeshurun and P. Lejay, *Phys. Rev. B* **44** 7167(R) (1991)
- [9] F.M. Sauerzopf, *Phys. Rev. B* **57** 10959 (1998)
- [10] T. Tamegai, T. Taen, H. Yagyuda, Y. Tsuchiya, S. Mohan, T. Taniguchi, Y. Nakajima, S. Okayasu, M. Sasase, H. Kitamura, T. Murakami, T. Kambara and Y. Kanai, *Supercond. Sci. Technol.* **25** 084008 (2012)
- [11] M. Eisterer, H.W. Weber, J. Jiang, J.D. Weiss, A. Yamamoto, A.A. Polyanskii, E.E. Hellstrom and D.C. Larbalestier. *Supercond. Sci. Technol.* **22** 065015 (2009)

- [12] M. Eisterer, M. Zehetmayer, H.W. Weber, J. Jiang, J.D. Weiss, A. Yamamoto and E.E. Hellstrom, *Supercond. Sci. Technol.* **22** 095011 (2009)
- 230 [13] L. Fang, Y. Jia, C. Chaparro, G. Sheet, H. Claus, M.A. Kirk, A.E. Koshelev, U. Welp, G.W. Crabtree, W.K. Kwok, S. Zhu, H.F. Hu, J.M. Zuo, H-H Wen and B. Shen, *Appl. Phys. Lett.* **101** 012601 (2012)
- [14] M. Shahbazi, X.L. Wang, S.R. Ghorbani, M. Ionescu, O.V. Shcherbakova, F.S. Wells, A.V. Pan, S.X. Dou and K.Y. Choi, *Supercond. Sci. Technol.* **26** 095014 (2013)
- 235 [15] M. Eisterer, V. Mishev, M. Zehetmayer, N.D. Zhigado, S. Katrych and J. Karpinski, *Supercond. Sci. Technol.* **27** 044009 (2014)
- [16] T. Taen, F. Ohtake, H. Akiyama, H. Inoue, Y. Sun, S. Pyon and T. Tamegai, *Phys. Rev. B* **88** 224514 (2013)
- 240 [17] T. Taen, Y. Nakajima, T. Tamegai, H. Kitamura and T. Murakami, *Physica C* **471** 784 (2011)
- [18] D.A. Moseley, K.A. Yates, N. Peng, D. Mandrus, A.S. Sefat, W.R. Branford and L.F. Cohen, *Phys. Rev. B* **91** 054512 (2015)
- [19] T. Taen, Y. Nakajima, T. Tamegai and H. Kitamura, *Phys. Rev. B* **86** 094527 (2012)
- 245 [20] M. Putti, M. Affronte, C. Ferdeghini, P. Manfrinetti, C. Tarantini and E. Lehmann, *Phys. Rev. Lett.* **96** 077003 (2006)
- [21] A.S. Sefat, R. Jin, M.A. McGuire, B.C. Sales, D.J. Singh and D. Mandrus, *Phys. Rev. Lett.* **101** 117004 (2008)
- 250 [22] C. Lester, J-H Chu, J.G. Analytis, S.C. Capelli, A.S. Erickson, C.L. Condon, M.F. Toney, I.R. Fisher and S.M. Hayden, *Phys. Rev. B* **79** 144523 (2009)

- [23] B. Rosenstein and V. Zhuravlev, *Phys. Rev. B* **76** 014507 (2007); B. Rosenstein and D. Li, *Rev. Mod. Phys.* **82** 109 (2010)
- 255 [24] Y. Nakajima, T. Taen, Y. Tsuchiya, T. Tamegai, H. Kitamura and T. Murakami, *Phys. Rev. B* **82** 220504(R) (2010)
- [25] N. Haberkorn, B. Maiorov, I.O. Usov, M. Weigand, W. Hirata, S. Miyasaka, S. Tajima, N. Chikumoto, K. Tanabe and L. Civalé, *Phys. Rev. B* **85** 014522 (2012)
- 260 [26] M. Nakajima, T. Liang, S. Ishida, Y. Tomioka, K. Kihou, C.H. Lee, A. Iyo, H. Eisaki, T. Kakeshita, T. Ito and S Uchida, *Proc. Natl. Acad. Sci. USA* **108** 12238 (2011)
- [27] T. Taen, F. Ohtake, H. Akiyama, H. Inoue, Y. Sun, S. Pyon, T. Tamegai and H. Kitamura, *Phys. Rev. B* **88** 224514 (2013)
- 265 [28] L.J. van der Pauw, *Philips Res. Rep.* **13** 1 (1958)
- [29] C.P. Bean, *Rev. Mod. Phys.* **36** 31 (1964)
- [30] P.A. Kes, J. Aarts, J. van den Berg, C.J. van der Beek and J.A. Mydosh, *Supercond. Sci. Technol.* **1** 242 (1989)
- [31] D. Dew-Hughes, *Philos. Mag.* **30** 293 (1974)
- 270 [32] R. Griessen, H-H Wen, A.J.J. van Dalen, B. Dam, J. Rector, H.G. Schnack, S. Libbrecht, E. Osquiguil and Y. Bruynseraede, *Phys. Rev. Lett* **72** 1910 (1994)
- [33] R.I. Rey, A. Ramos-Ivarez, C. Carballeira, J. Mosqueira, F. Vidal, S. Salem-Sugui, A.D. Alvarenga, Rui Zhang and H-Q Luo, *Supercond. Sci. Technol.* **27** 075001 (2014); S. Salem-Sugui, A.D. Alvarenga, R.I. Rey, J. Mosqueira, H-Q Luo and X-Y Lu, *Supercond. Sci. Technol.* **26** 125019 (2013)
- 275